# **X 5** Synthesis of Molecular Magnets

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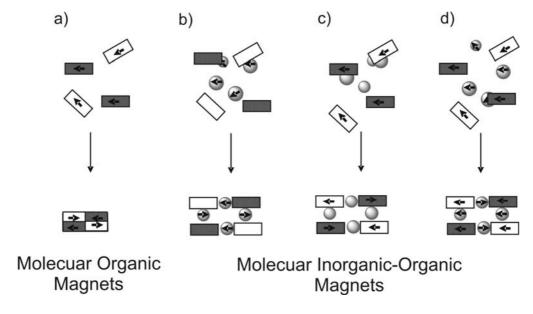
#### 1 Introduction

Molecular magnetism can be defined as the interdisciplinary area where the techniques of molecular chemistry are exploited in order to design and to synthesize new classes of magnetic materials, which are based on molecular lattices. The constituent molecular bricks have to bear unpaired electrons, which can be located either in s- or p-orbitals, as the case in pure organic magnets, or in d or f-orbitals involving transition, lanthanide or even actinide metal ions.

In comparison to nanoparticles of similar size, molecular compounds exhibit *a priori* high monodispersity in size, volume, shape and charge. Additionally, they are soluble in common solvents and the modular character opens ways to fine-tune their properties. With regard to the magnetic properties, such molecular systems exhibit features which might be considered both in classical as well as in quantum regime.

The field of molecule based magnetism involves materials built-up from (i) pure organic bricks or (ii) from the combination of organic material and inorganic metal ions or clusters (Figure 1). The utilisation of metal ions, by essence has capitalized on the specific properties of structurally well-defined and versatile supramolecular materials. [1] Resulting supramolecular motifs are one-, two- and three dimensional network architectures, which were self-assembled from (i) polyatomic, organic radical-bearing ligands and (ii) metallic centres that carry a magnetic moment. [2]

By combination of organic synthesis with the supramolecular structuring approach, any deliberate arrangement of paramagnetic units in space is feasible within an extended network, from which the magnetic properties will spring up as consequence of specific short and long range ordering phenomena.



**Fig. 1:** Schematic representation of the self-assembly of molecular building blocks to magnetic materials: Self-assembly of a) only organic molecules (rectangles) or b-d) together with inorganic ions or clusters (spheres); the arrows mark the spin-bearing constituents, respectively.

### 2 Molecular Organic Magnets

Until the beginning of the 1990es, no genuine organic magnetic material only based on s- and p- electrons was known. The pioneering work of Itoh and Iwamura [3] showed for the first time that strong ferromagnetic coupling could be realised in polycarbenes (Figure 2, top). The two unpaired electrons on each centre are in orthogonal orbitals, therefore they are in a S=1 state. The disposition in *meta* around the aromatic rings results in quasi degenerate molecular orbitals, with strong ferromagnetic coupling.

**Fig. 2:** Two examples of molecular bricks forming purely organic molecular magnets with high temperature magnetic ordering: polycarbenes (top) and nitronyl nitroxide radicals (bottom)

In general, there three conditions necessary to achieve a purely organic, preferably high, temperature magnets: (i) assemblage of organic-based unpaired electrons in high concentration; (ii) operation of strong exchange interactions aligning the spins; and (iii) formation of magnetic domains in which all the spins are ordered in extended networks in mesoscopic scale. However, the realization of necessarily strong exchange coupling between neighbouring organic molecules through van der Waals forces, Hydrogen bonds, hydrophobic interaction, etc. is expected to be difficult. Nevertheless, relatively strong ferromagnetic ordering was found for the class of organic nitronyl nitroxide radicals (Figure 2, bottom) with the, so far, highest reported ordering temperature of 1.4 K. [4] The use of fullerenes raised the ordering temperature as expression of the coupling strength up to 16 K, [5] while by using sulphur nitrogen radicals eventually 36 K could be achieved. [6]

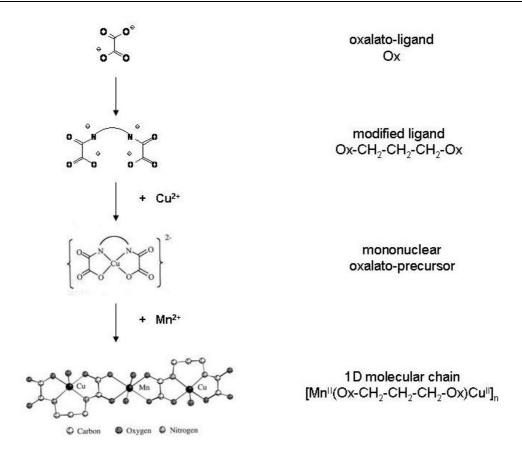
## 3 Molecular Inorganic-Organic Magnets

#### 3.1. Infinite 1D-, 2D-, and 3D-Networks: Molecular Magnetic Solids

#### 3.1.1. Oxalato-based Magnets

One of the challenges in the field of supramolecular self-assembly of molecular inorganic-organic magnets is the controlled, spontaneous generation of well-defined architectures from both organic and inorganic building blocks (see Figure 1, b-d). A high structural organization can be ensured through the multiple one- to three-binding of transition metals giving rise to a variety of extended networks. The emphasis here lies in the design of the organic bricks (thereafter called ligands) together with the proper choice of the metal ions, since the overall topology of the network is strongly influenced by the match (or mismatch) of coordination algorithm of the linking metal ions with the binding properties (electronic, spatial, etc.) of the bridging ligands. Thus, the *bis*-chelating coordinating ability of ligands of the oxalate-type (Ox) has made them to a versatile choice for interconnecting different spin-bearing metal ions to extended networks (Figure 3, top). Indeed, the self-assembly of inorganic architectures incorporating bridging oxalato-type units has yielded various types of frameworks, which include 1D-chains, [7] 2D-layers, [8] and 3D-networks. [9]

Most of the so far investigated infinite chain-like compounds use the modified oxolato ligand Ox-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-Ox with two interconnected oxalato-units and are of bimetallic character as shown for compound [Mn<sup>II</sup>(Ox-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-Ox)Cu<sup>II</sup>(Solvent)<sub>x</sub>]<sub>n</sub> in Figure 3, bottom. [10] The magnetic properties between 5 and 300 K can be interpreted as an antiferrimagnetic coupling between the two neighbouring spins (here  $S_{Mn} = 5/2$  and  $S_{Cu} = \frac{1}{2}$ ), while below 5 K magnetic AF ordering due to interchain exchange can be observed. [11] All investigated compounds of this type exhibit antiferrimagnetic ordering at low temperatures with  $T_N$  between 1.8 and 5 K, only the compound [Mn<sup>II</sup>(H<sub>2</sub>O)<sub>2</sub>(Ox-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-Ox)Cu<sup>II</sup>(H<sub>2</sub>O)]<sub>n</sub> incorporating an OH-substituted interconnecting linker exhibits ferrimagnetic ordering with  $T_N = 4.6$  K. The occurrence of both anti- and ferrimagnetic ordering can be explained as a combination of intrachain and interchain ligand-mediated metal-metal exchange interactions. Additionally, some of these compounds behave as metamagnets, where a small applied magnetic field (between 1-2 kOe) induces a long-range ferromagnetic-like ordering by compensating the rather weak (antiferromagnetic) interaction between the 1D-chains. [12]



**Fig. 3:** Reaction scheme for the formation of the 1D oxalato-based molecular compound [Mn<sup>II</sup>(H<sub>2</sub>O)<sub>2</sub>(Ox-CH<sub>2</sub>-CH<sub>2</sub>-CH<sub>2</sub>-Ox)Cu<sup>II</sup>(H<sub>2</sub>O)]<sub>n</sub>.2H<sub>2</sub>O exhibiting intrachain antiferrimagentic coupling. [10]

In order to increase the magnitude of the magnetic ordering, the Ox-ligands have been equally used in achieving 2D-spatial arrangement of spin-bearing metal-ions. Thus, the molecular building blocks  $[M^{II}M^{III}(Ox)_3]^{1-}$  and  $[M^{II}(Ox-CH_2-CH_2-CH_2-Ox)_3Cu^{II}_3]^{2-}$  (with M=1 transition or lanthanide metals) can be ordered in sheet-like architectures, whereby the resulting negative charge of the resulting 2D network structure is counterbalanced by positively charged cations  $A^+$ . [13] [14] The nature of the electrostatically and interstitially bound cations  $A^+$  determines, together with the chirality of coordinated metal centres, the overall structure by discriminating from other isoconstitutional and isostoichiometric 3D-networks. Thus, two dimeric units of alternating chirality are necessary to form a closed planar 2D structure with a hexagonal network motif. [15] The cations  $A^+$  are mostly located between the anionic honeycomblayers and the systematic variation of  $A^+$  can be used to modulate the separation between the layers and hence the interlayer magnetic exchange interaction.

Such Ox-based 2D polymeric materials exhibit a wide range of magnetic behaviour, and ferro, ferri or antiferromagnetic long-range ordering processes with critical temperatures ranging from 5-44 K, as well as short-range correlations and spin glass like behaviour have all been observed. Generally, the magnetic interactions and magnetic ordering in  $[A][M^{II}M^{II}(Ox)_3]$  compounds is strongly dependant on the character of the cations  $A^+$  as well as the combination of  $M^{II}$  and  $M^{III}$  metal ions.



**Fig. 4:** Molecular structure of the hybrid material [BEDT-TTF]<sub>3</sub> [Mn<sup>II</sup>Cr<sup>III</sup>(Ox)<sub>3</sub>]<sub>n</sub>: Top-view showing the oxolato-based bimetallic [Mn<sup>II</sup>Cr<sup>III</sup>(Ox)<sub>3</sub>] honeycombe-like sublattice (left) and side-view with representation of the alternation of the organic [BEDT-TTF]-and of the inorganic [Mn<sup>II</sup>Cr<sup>III</sup>(Ox)<sub>3</sub>] layers (centre). Thermal variation of the electrical resistivity with the inset showing the low-temperature magnetoresistance with the magnetic field applied perpendicular to the layers (right). [16]

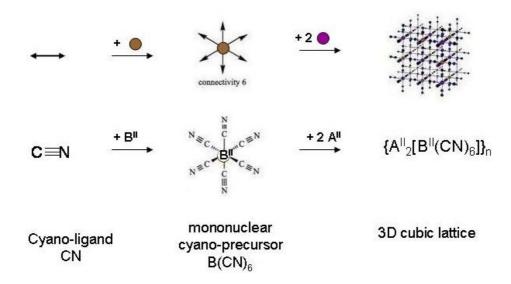
The highlight of this anionic cation A<sup>+</sup>/2D-sheet strategy has been the realisation of the coexistence of ferromagnetism and metallic conductivity in the same material. The compound [BEDT-TTF]<sub>3</sub>[Mn<sup>II</sup>Cr<sup>III</sup>(Ox)<sub>3</sub>] is built up from two molecular sublattices: a bimetallic 2D honeycomb network with ferromagnetically coupled metal ions (T<sub>c</sub> = 5.5 K) and layered BEDT-TTF cations, which are intercalated between the honeycombe-like oxalate layers and introduce electric conduction into the magnetic molecular architecture (Figure 4). [16] The self-assembly of Ox-bridged bimetallic units containing equal chiral information in both metallic centres results necessarily in the formation of 3D network structures. [15] The possibility of linking paramagnetic metal ions via Ox-bridges in 3D lattices opens interesting avenues to tailor-made functional materials. In recent years, the group of S. Decurtins reported on the structure of several anionic 3D networks based on the Ox-system  $[M^IM^{III}(Ox)_3]^{2-}(A^+)_2$  or  $[M^I_2(Ox)_3]^{2-}(A^+)_2$  ( $M^{II/III}$  transition metals and  $M^I = Li$ , Na), where the cationic part  $A^+$  consists of transition metal imino complexes  $[M^{II}$  (bpy)<sub>3</sub>]<sup>2+</sup> (with M = Fe, Ru, Ni, Co), which can be used as kind of "Trojan Horse" to introduce molecular functionalities into the material. All so far investigated compounds exhibit an analogous anionic chiral 3D (10,3) network, whereby the ten-gon framework demonstrates a marked structural flexibility towards the inclusion of the cations A<sup>+</sup> into the internal cavities (Figure 5). [17] The robustness of the oxolato approach was also demonstrated by the isosteric replacement of the oxolato bridges by dithioxolato bridges (C<sub>2</sub>O<sub>2</sub>S<sub>2</sub><sup>2</sup>-), what equally resulted in a topology, which is constituent to the chiral 3D (10.3) network. Finally, enantiomeric pure 3D network  $[Cr^{III}(Ox)_3]^{3-}$  polymers could be obtained by the introduction of homochiral, that means enantiomeric resolved [Ru<sup>II</sup>(bpy)<sub>3</sub>]<sup>2+</sup>, cationic A<sup>+</sup> building blocks. Such homochiral materials are perspective to exhibit very interesting magneto-optical effects. [18]

**Fig. 5:** Schematic representation of the stepwise formation of the bimetallic 3D anionic (10,3) network formed from the oxalato ligand Ox. [15]

The 3D homo- and heterometallic oxolato bridged frameworks are examples of supramolecular compounds, which show long-range magnetic ordering and, additionally, the possibility to generate further physical properties introduction of guests  $A^+$  into the internal cavities (Figure 5). Magnetic studies of the Ox-based 3D inclusion compound  $[Co^{III}(bpy)_3][Co^{II}_2(Ox)_3]$  ClO<sub>4</sub> show the occurrence of a weak ferromagnetism at low temperatures (Tc = 9.0 K). This magnetic ordering is attributed to the spin canting of the magnetic moments in the chiral 3D oxalato-bridged cobalt (II) network and it was shown that the magnetic ordering strongly depends on the size and the magnetic character of the trischelated counterions  $[M^{III}(bpy)_3]^{2+}$ . [19]

#### 3.1.2. Cyano-based Magnets

The rational built-up approach of bimetallic cyano compounds is based on (i) the linear bidentate binding mode of the cyan-bridging ligand (thereafter CN) and on (ii) the six-fold connectivity of the  $B^{II}(CN)_6$  brick (B= first row transition metal), which should result in combination with octahedral coordinating metal ions  $A^{II}$  in a neutral infinite cubic crystal lattice (Figure 6).



**Fig. 6:** Schematic representation of the self-assembly process of a infinite 3D cubic coordination network out of the CN ligand and two types of different metal ions B (brown) and A (violet).

The use of  $A^{III}$  metal ions will result in (ideally) face-centred cubic negatively charged network structure  $\{A^{III}_{p}[B^{II}(CN)_{6}]_{q}\}^{1}$ , in which the electroneutrality is established by the introduction of monovalent alkali ions  $C^{1+}$  (C=Na, K, Rb, Cs) at interstitial lattice positions. This simple and appealing cubic structure is well-known as "Prussian blue" structure, but is also strongly reminiscent to the lattice structure of perovskites. Chemical synthesis can control the nature of the involved metal ions, but can also create sizable departures from the ideal stoichiometry with  $C^+$  vacancies or  $B(CN)_6$  substitutions partially filled with solvent molecules. The very general cubic A-CN-B matrix offers a wide variety to develop the orbital interactions in the three directions of space by choosing very different paramagnetic anions A and B. The high symmetry of the system allows an efficient control of the nature and the amplitude of the magnetic exchange interactions between the metal ions.

It was very early already shown that the original mixed valence Prussian-blue compound (A =  $Fe^{III}$  and B =  $Fe^{II}$ ) exhibits long-range ferromagnetic ordering at  $T_c = 5.6$  K. [20] This is astonishing, since only the  $Fe^{III}$  sites carry a spin, whereas the  $Fe^{II}$  sites are diamagnetic and, consequently, the magnetic interactions must occur between the next-nearest ions through  $Fe^{III}$ -C-N- $Fe^{II}$ —N-C- $Fe^{III}$  linkages over a distance of 10.6 Å. This illustrates the possibility of propagating magnetic interactions through widely separated spin carriers in molecular compounds mainly due to the strong spin delocalization from the metal ions towards its nearest neighbours. The presence of strong spin densities on the nitrogen and carbon atoms of the cyano groups has been experimentally observed by polarized neutron diffraction. [21] Depending on the nature of A and B, one may obtain 3D anti-and ferromagnets or and ferrimagnets. In Prusssian-blue-like phases, the metal sites A and B are in octahedral surroundings, where the 3d valence orbitals are split into low-lying ( $t_{2g}$ ) and high-lying ( $e_g$ ) subsets. For B only, the low-lying  $t_{2g}$  orbitals are occupied, so the magnetism shown by the overall compound depends mainly by the nature of metal ion A.

If all unpaired electrons of A occupy  $t_{2g}$ -orbitals, the A-B interaction will be antiferromagnetic, and the compound will be a ferrimagnet, provided there is no compensation of the spin moments. If the unpaired electrons of A occupy some  $t_{2g}$  orbitals

and some  $e_g$ -orbitals, the A-B interaction will again be antiferromagnetic, but less pronounced than in the previous case, and the compound will be a ferrimagnet with a lower critical temperature. Finally, if the unpaired electrons electrons of A only occupy  $e_g$ -orbitals, the A-B interaction will be ferromagnetic, and the compound will be a ferromagnet.

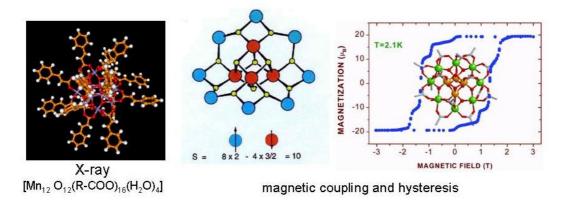
Moving along this rationale, several molecular-based 3D solids with astonishing magnetic properties could be synthesized during the last decade. Thus, the first highlight was the synthesis of the ferrimagnet  $\text{Cr}^{\text{II}}_{3}[\text{Cr}^{\text{III}}(\text{CN})_{6}]_{2}$ . 10 H<sub>2</sub>O exhibiting a  $T_{C}$  of 230 K, which was followed by the synthesis of the compound  $V^{\text{II}}_{0.42}V^{\text{III}}_{0.58}[\text{Cr}^{\text{III}}(\text{CN})_{6}]_{0.86}$ . 2.8 H<sub>2</sub>O being a room temperature molecular magnet with a  $T_{c}$  of 315 K. [22] [23] The  $V^{\text{II}}$  metal ions were partially oxidized to  $V^{\text{III}}$ , however, the magnetic orbitals of all involved metal ions ( $V^{\text{II}}$ ,  $V^{\text{III}}$  and  $V^{\text{III}}$ -Cr interactions antiferrimagnetically, the overall magnetic coupling scheme results in a ferrimagnet with very high critical temperature.

However, the stoichiometric composition of the compound  $V^{II}_{0.42}V^{III}_{0.58}[Cr^{III}(CN)_6]_{0.86}$ . 2.8  $H_2O$  points rather to an amorphous and non-stoichiometric extended solid than a defined molecular compound. In this respect, this system resembles to the first, previously reported room temperature magnet synthesized from molecular precursors. Although its structure is still under discussion, the amorphous material of the composition  $V^{II}(TCNE)_2$ . 1/2  $(CH_2Cl_2)$  (TCNE being the radical anion of tetracyanoethylene) is even exhibiting an (estimated)  $T_C$  of 350 K. [24]

Furthermore, Hashimoto et al. observed interesting photo-induced magnetization phenomena in a powder of the stoichiometry of  $K_{0.2}Co_{1.4}Fe(CN)_6$ . 6.9  $H_2O$ . Its magnetisation and  $T_c$  values are significantly increasing by red light illumination and the enhanced magnetization is partly reduced by blue light illumination. This photo effect is explained by the charge-transfer induced spin transition between  $Fe^{II}$  ( $t_{2g}^{\ 6}e_{g}^{\ 0}$ ; S=0)-CN-Co<sup>III</sup>( $t_{2g}^{\ 6}e_{g}^{\ 0}$ ; S=0) and  $Se^{III}$  ( $Se^{\ 5}e_{g}^{\ 0}$ ) states by light irradiation. [25]

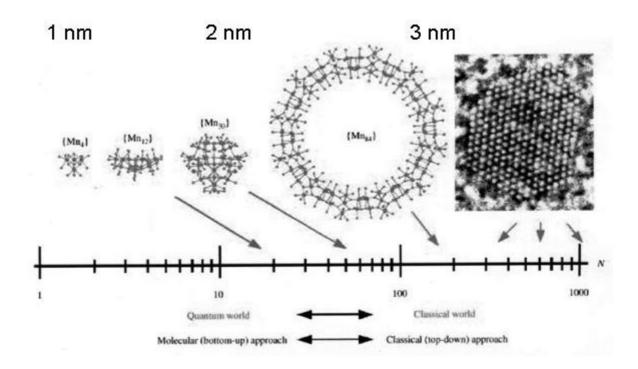
#### 3.2. Finite Cluster Compounds: Single-Molecule Magnets (SMMs)

An exciting development in nanoscale magnetic materials occurred in 1993 when [Mn<sub>12</sub>O<sub>12</sub>(O<sub>2</sub>CMe)<sub>16</sub>(H<sub>2</sub>O)<sub>4</sub>] (hereafter Mn<sub>12</sub>) was identified as nanoscale magnet, [26] the first to comprise secrete, magnetically non-interacting molecular units rather than a 3D extended lattice (as in metals and metal oxides, for example). The discovery initiated the field of molecular nanomagnetism and such molecules have since been termed single-molecule magnets (SMMs). They derive their special magnetic properties from the combination of a large spin (S) and an Ising (easy-axis) magneto-anisotropy exhibiting a negative zero-field splitting parameter (D), which gives rise to the superparamagnetic–like property of a barrier to magnetization relaxation (Figure 7). This class of compounds does not only display magnetisation hysteresis, but also quantum tunnelling of magnetisation (QTM) [27] and quantum phase interference. [28]



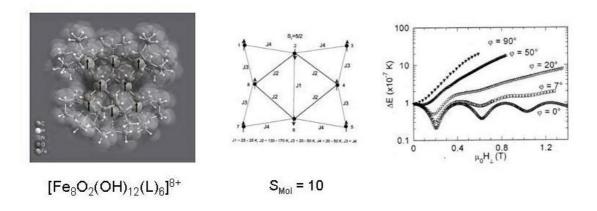
**Fig. 7:** Molecular structure (left), magnetic coupling scheme and hysteresis (right) of the  $Mn_{12}$  family exhibiting magnetic hysteresis and quantum tunnelling of magnetization (OTM).

Since the initial discovery of  $Mn_{12}$  molecule, a number of other types of molecules have been discovered, almost all of them being transition metal clusters, and the majority of them being Mn-oxo clusters containing 2-84 Mn<sup>III</sup>, Mn<sup>III</sup> and Mn<sup>IV</sup> ions, a multitude of oxo bridges (M-O-M), and several peripheric organic, mostly carboxylate, ligands. [29] So far, the synthetic efforts towards new compounds of this family were guided along two main lines: (i) to control the nuclearity of the cluster systems and (ii) to increase the anoisotropy barrier height of the intrinsic magnetism. The combined synthetic and magnetic efforts mainly of the groups of Wernsdorfer and Christou yielded a large variety of  $Mn_x$ -cluster SMMs exhibiting spherical, disk-like, ring-like, or even rod-like morphologies and topologies (Figure 8). [30]



**Fig. 8:** Several examples of Manganese-oxo-cluster SMMs depicted together with their size (top) and Néel-vektors (bottom) in comparison to a superparamagnetic nanoparticle (right). (taken from [29]).

A further group of SMMs intensively investigated during the last years is the  $Mn_4O_3X$  family, in which the manganese ions are exhibiting a distorted-cubane core of virtual  $C_{3\nu}$  symmetry. The metal centres are mixed valence exhibiting frequently three  $Mn^{III}$  and one  $Mn^{IV}$ , which are coupled by  $\mu_3$ -O-bridges to a well isolated ground state spin of S = 9/2. [31] Some of these complexes crystallize as head-to head dimmers, which have allowed for the first time the identification of exchange-biased quantum tunnelling of magnetization (QTM) [32] and of quantum coherence in SMMs. [33]



**Fig. 9:** Molecular Structure (left), magnetic coupling scheme (centre), and experimental oscillation of the tunnel splitting (right) of [{Fe<sup>III</sup><sub>8</sub>O<sub>2</sub>(OH)<sub>12</sub>(L)<sub>6</sub>}Br<sub>7</sub>.H<sub>2</sub>O]Br.8H<sub>2</sub>O (L= organic ligand). [34]

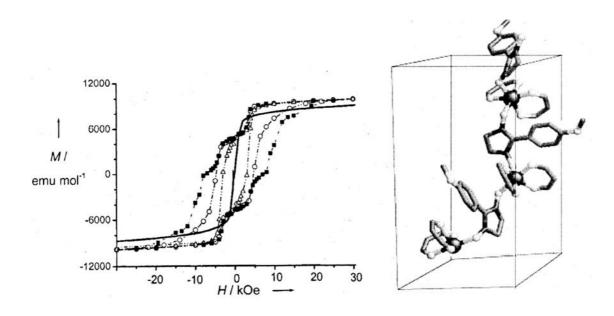
Tetranuclear  $Fe^{II}_{4}$  cuban-like and nononuclear  $Fe^{II}_{9}$  complexes were synthesized recently showing only moderate magnetic anisotropy. [35] [36] One of most intensely studied SMMs represents the molecule [ $\{Fe^{III}_{8}O_{2}(OH)_{12}(L)_{6}\}Br_{7}.H_{2}O]Br_{8}H_{2}O$  (L= organic ligand), [37] in which for the first time oscillation of the quantum tunnel splitting could be observed (Figure 9). [34] Furthermore, several extended  $Fe^{III}_{19}$  oxo-cluster were prepared exhibiting SMM behaviour with modest barrier heights. [38] Finally, various SMMs incorporating different types of transition metals of various nuclearity  $M_x$  (with  $M = V^{III}$  [39],  $Co^{II}$  [40],  $Ni^{II}$  [41]), including mixed-metal systems, [42] have been prepared with S values ranging with 3 to 13.

### 3.3. Infinite 1D Complexes: Molecular Single Chain Magnets (SCM's)

Taking into account the above described characteristics of SMMs (high spin ground state S, negative zero field splitting parameter D with resulting magnetic anisotropy axis), one idea would be to go one step further towards extended 1D restricted molecular systems and to synthesize magnetically active infinite coordination polymers, which have been named "single chain magnets" (SCM). To design this type of system, three essential ingredients need to be considered: (i) Spin carriers must exhibit a strong uniaxial anisotropy to be able to block or to "freeze" their magnetization in one direction. (ii) The material needs to exhibit a spontaneous magnetization to be called a magnet, whereby the individual magnetic moments in the chain must not cancel out in the "frozen" state. (iii) The chains must be isolated magnetically as much as possible to avoid 3D ordering. The occurrence of slow relaxation and Arrhenius behaviour of the relaxation time in such systems was already predicted theoretically by Glauber using a 1D Ising model in the 1960ies. [43]

On the base of Glauber's prediction, helical Co<sup>II</sup>-nitronyl nitroxide chains [[Co<sup>II</sup>(hfac)<sub>2</sub>(NITPhOMe); Figure 10) acting as molecular magnetic wires were first synthesized and investigated by Gatteschi et. al. in 2001. [44] The molecular chains behave as 1D ferrimagnets, because of the non-compensation of the magnetic moments of the Co<sup>II</sup> and the organic radical centres. The susceptibility below 50 K is strongly anisotropic with an easy

axis of magnetization and the relaxation time of the magnetization shows strong frequency dependence in ac susceptibility measurements. This seems to be physically reasonable, since the Glauber model for Ising ferromagnetic chains predicts that the reversal of the spins becomes more difficult on lowering the temperature because the short-range correlation along the chains favours the parallel alignment of the spins. Consequently, an Arrhenius behaviour is predicted for the relaxation time, with the height of the barrier corresponding to the nearest-neighbour exchange interaction. [45]



**Fig.10:**Molecular structure of the helical Co<sup>II</sup>-nitronyl nitroxide chain [[Co<sup>II</sup>(hfac)<sub>2</sub>(NITPhOMe) and hysteresis loops observed on a single crystal by applying magnetic fields parallel to the chain axis at 2.0 K (squares), 3.0 K (circles), 4.5 K (triangles), and at 2.0 K parallel to it (line). [44]

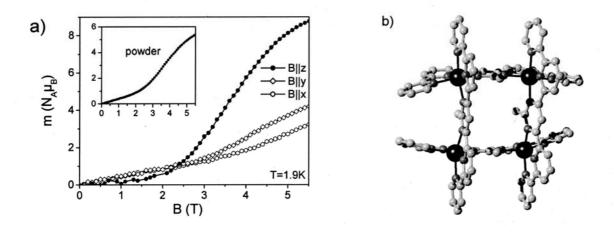
Recently, the synthesis and magnetic investigation of two further, now mixed-metal, 1D molecular chains,  $[\{Fe^{III}(L)(CN)_4\}_2Co^{II}(H_2O)_2].4H_2O$  and  $[Mn_2(L^1)_2Ni(L^2)(Py)_2]$  (ClO<sub>4</sub>)<sub>2</sub>, confirmed principally the validity of the Glauber approach to achieve SCMs. [45] [46] Interestingly, other, similar 1D systems containing Mn<sup>II</sup> and Mn<sup>III</sup> metal ions did not show the equivalent magnetic behaviour.[47] [48]

#### 3.4. Magnetic Molecular Wheels

Molecular wheels are excelled by a virtually perfect ring-like arrangement of the metal ions within a single molecule. The decanuclear wheel [Fe<sup>III</sup><sub>10</sub>(OCH<sub>3</sub>)<sub>20</sub>(O<sub>2</sub>CCH<sub>2</sub>Cl)<sub>10</sub>], generally called Fe<sub>10</sub>, has become the prototype for this class of compounds. [49] In the last decade, various wheels with different metal ions and number of centres were realized, whereby a difficult to rationalize preference of even-numbered wheels has been manifested. [50] In view of their structure, antiferro- or ferromagnetically coupled wheels are widely regarded as finite equivalents for infinite magnetic 1D chains (*vide infra*), implying that physical

concepts found for these chains should describe them, too. Even-numbered, AF-coupled wheels are characterized by an S=0 spin ground state, which makes them unsuitable as a SMMs. However, excited magnetic states seem to be easily accessible and first insight into the higher lying magnetic spectrum was given by the observation of steps in the magnetization and torque curves at very low temperatures. [51] Additionally, the molecular wheels might show large quantum effects, such as tunnelling of the Néel vector [52] or quantum coherence phenomena. [53] Furthermore, very similar wheels systems form a suitable series of objects to investigate the origins of magnetic anisotropy, partly because of their high symmetry, which was investigated in closer detail in different Co<sup>II</sup><sub>4</sub> and Cr<sup>III</sup><sub>8</sub> systems. [54] [55]

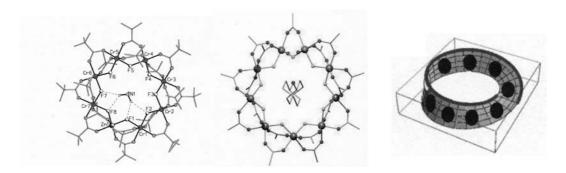
Magnetic studies of the Co<sup>II</sup><sub>4</sub> systems as function of temperature revealed intramolecular antiferromagnetic exchange interaction. [56] Subsequently recorded magnetisation curves on single crystals of certain Co<sup>II</sup><sub>4</sub>- rings at 1.9 K unravelled following anisotropic situation: For magnetic fields perpendicular to the grid plane, the magnetization shows a behaviour reminiscent of a thermally broadened magnetization step due to a ground-state level crossing, as it is often observed in antiferromagnetic clusters. [57] In contrast, the magnetic moments for applied fields within the plane of the Co<sup>II</sup><sub>4</sub>-wheel, increase linearly with equal slope as function of the magnetic field (Figure 11). No longe-range ordering develops as intermolecular interactions are negligibly small preventing the system from undergoing a metamagnetic phase transition. However, the analysis of its magnetism can be completely recast in the language of metamagnetism, and in this sense it was proposed to call them examples for single-molecule metamagnets. [58]



**Fig. 11:** Molecular structure Co<sup>II</sup><sub>4</sub> wheels (right) and the field dependence of the magnetic moment of a single crystal at 1.9 K for magnetic fields along the main axes (left). The inset displays the magnetization curve of the powder sample. [58]

Ways to avoid the diamagnetic S=0 ground state consist in either (i) the introduction of hetero-metal ion into the ring or (i) the synthesis of odd-numbered rings. Both approaches will lead to wheels, which should show magnetic frustration and examples proving the feasibility of both ways could be realized: Following the general protocol of the formerly described  $Cr^{III}_8$  wheel, a series of hetero-metal ion wheels of the composition  $Cr^{III}_7M$  (with  $M = Ni^{II}$ ,  $Co^{II}$ ,  $Mn^{II}$ , and the diamagnetic  $Cd^{II}$ ) and  $Cr^{III}M_2$  (with M = V=0) could be obtained.

[59] First magnetic investigations have proven, that the Cr<sup>III</sup><sub>7</sub>M wheels act, as expected, as antiferromagnetically coupled wheels with a nondiamagnetic ground state at low temperature.

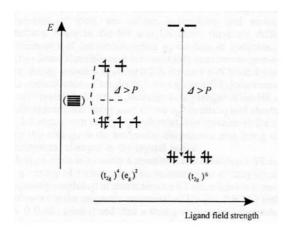


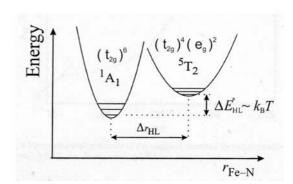
**Fig. 12:** Schematic molecular structures of the heterometallic, even-numberd  $Cr^{III}_{7}Zn^{II}$  wheel (left) and the hetero-metalli, odd-numbered  $Cr^{III}_{8}Ni^{II}$  wheel (centre). right) The visualisation of the Möbius-strip type "magnetic knot" delocalized on the  $Cr^{III}_{8}$  chain (black circles, white circle:  $Ni^{II}$ ). [60]

Using the same synthetic procedure, odd-numbered  $Cr^{III}_8Ni^{II}$ ,  $Cr^{III}_8Co^{II}$  and  $Cr^{III}_7(V=O)_2$  were accessible. The magnetic behaviour of these nonanuclear wheels is fascinating. The  $Cr^{III}_8Ni^{II}$  compound contains an even number of unpaired electrons due to the presence of a  $Ni^{II}$  ion, with S=1, and eight  $Cr^{III}$  ions with S=3/2 spins; this would even allow a diamagnetic ground state. Nevertheless, not all the antiferromagnetic interactions can be simultaneously satisfied within the  $Cr^{III}_8Ni^{II}$  wheel and therefore it can be regarded as frustrated. A way to visualize this problem is its analogy with the Möbius strip having a "magnetic knot", which represents within the wheel the point of spin frustration (where neighboured spins can not align antiparallel). Detailed magnetic investigations could determine that this "magnetic knot" is clearly delocalized within the  $Cr^{III}_8$  chain and not at the single  $Ni^{II}$  ion (Figure 12). [60]

### **4** Spin Transition (ST) Compounds

As a consequence of the splitting of the energy of the d-orbitals into  $t_{2g}$  and  $e_g$  sets in a ligand field, octahedral complexes of certain transition metal ions, particularly those of the first transition metal series with configurations of  $d^4$  to  $d^7$ , may exist in either high-spin (HS) or low-spin (LS) state, depending on the nature of the ligand field around the metal ion. In weak fields the ground state is HS where the spin multiplicity is a maximum, whereas the strong ligand fields stabilize the LS state with minimum multiplicity. For intermediate fields, the energy difference ( $\Delta E^0_{HL}$ ) between the lowest vibronic levels of the potential wells of two states may be sufficiently small such application of some relatively minor external perturbation effects a change in the state. This phenomenon is known as a spin transition (ST) and it might be induced thermally if  $\Delta E^0_{HL} = k_B T$ , but also pressure-, light-, and soft x-ray induced spin transition phenomena have been observed. [61] [62] [63]



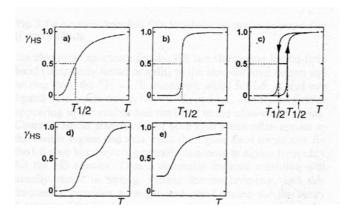


**Fig. 13:** Representation of the two possible electronic configurations of a d<sup>6</sup> system surrounded by an octahedral ligand field (∆: ligand field parameter; P: spin pairing energy; left). The resulting potential wells for the diamagnetic <sup>1</sup>A<sub>1</sub> and the paramagnetic <sup>5</sup>T<sub>2</sub> states (here for octahedral Fe<sup>II</sup>), the nuclear coordinate being the metal-donor atom distance (right). [64]

Although spin transitions have been observed for all the above mentioned electron configurations of the first transition series, by far the greatest number of research work is dealing with  $Fe^{II}$  and some  $Co^{II}$  complexes. Especially ST-Fe<sup>II</sup> compounds are magnetically a very appealing research object, since the spin transition occurs between the diamagnetic  $^{I}A_{I}$  (LS) and the paramagnetic  $^{5}T_{2}$  (HS) states (Figure 13), what can be easily followed by magnetic measurement techniques.

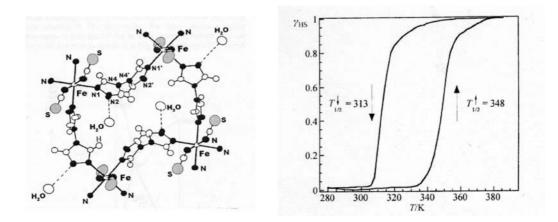
The driving force of the thermally induced spin transition is the considerable entropy gain (50-80 J mol<sup>-1</sup> K<sup>-1</sup> for Fe<sup>II</sup>) for the LS to HS conversion, made up of a magnetic component  $\Delta S_{\text{Mag}}$  and a much greater vibronic entropy contribution  $\Delta S_{\text{Vib}}$  arising from the higher degeneracies (degrees of freedom) in the HS state. The enthalpy change associated with the LS to HS conversion is typically in the range 6-15 kJ mol<sup>-1</sup> (for Fe<sup>II</sup>) and is primarily due to

the rearrangement of the coordination sphere (e.g. metal donor atom bond length increase). [65] In the solid state, there is electron-phonon coupling in order to communicate the spin state change between different molecules within the lattice. It is proposed that a local entropy driven spin state change accompanied with a large change in the metal-donor atom bond distance (e.g. 0.2 A for Fe<sup>II</sup>-N) creates a "point defect", which is communicated to the surroundings *via* phonon interactions and resulting in a cooperative spin state change. [66] From the measurement of the temperature-dependence properties of a ST system it is possible to evaluate the relative concentrations of HS and LS states in function of temperature. Regarding the plot of the HS molar fraction  $\gamma_{HS}$  vs. T, different forms of transition expressing different degrees of cooperativity can be found (Figure 14 a-e). The spin transition might be gradual and continuous over an extended temperature range as found in solutions, or it may be abrupt and occur within a narrow temperature range. The spin transition might be associated with a thermal hysteresis loop, or be a multi step process. In certain cases the spin transition might be incomplete at one or both extremes of the curve. The curves are diagnostic of the nature of the ST and the steepness of change is indicative, at least in solid state, of the extent of cooperativity involved into the propagation of spin change throughout the crystal lattice. A transition temperature is defined as that temperature at which fractions of HS and LS species, which take part in transition, are equal.



**Fig. 14:** The nature of ST curves; a) gradual; b) abrupt; c) with hysteresis; d) with steps; e) incomplete. [64]

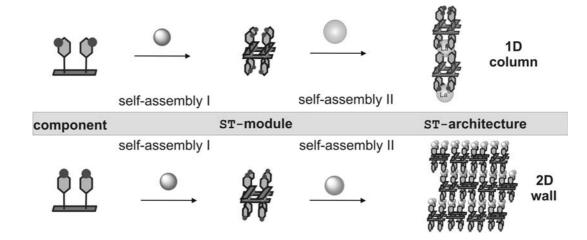
By far most of the so far investigated complexes exhibiting ST (>150) are mono-nuclear Fe<sup>II</sup> compounds, where the metal ion is surrounded by a six N-donor atoms. The donor atom set is mostly part of a heterocyclic ring systems, which can be rigidly arrange around the metal ion and allow fine-tuning of the ligand field strength by (i) introduction of sterically crowding substituents close to the donor atoms; (ii) by changing the ring size from six to five membered rings, what reduces both the  $\sigma$ - and the  $\pi$ -acceptor strength of the ligands, and (iii) selective replacement of N-donor-atoms by less strongly coordinating O- or S- donor atoms. The first synthetic system, [Fe (phen)<sub>2</sub>(NCS)<sub>2</sub>] (phen = 1,10 phenantroline) was synthesized in 1966 and since then many other compounds of similar composition have been shown to undergo transitions (one typical example of a mononuclear ST- Fe<sup>II</sup>-complex shown in Figure 15). [67]



**Fig. 15:** Molecular structure (left) and temperature-dependence of the HS molar fraction (right) of the mononuclear spin transition complex [Fe(NH<sub>2</sub>trz)<sub>2</sub>] (NO<sub>3</sub>)<sub>2</sub>. [68]

More recently, several binuclear ST-Fe<sup>II</sup><sub>2</sub> complexes and even a binuclear ST-Co<sup>II</sup><sub>2</sub> could be synthesized. [68] [69] Furthermore, a series of tetranuclear ST-Fe<sup>II</sup><sub>4</sub>-complexes was reported as so far largest oligomeric molecular architectures undergoing ST. [70] Interestingly, in both bi- and tetranuclear species, covalently linking of Fe<sup>II</sup> spin transition centres leads to very broad and gradual transitions without any hysteresis phenomena, which results in final consequence to a less discriminating magnetic signal and loss of bistability. This particular behaviour is assigned to the presence of a statistic distributions of magnetically [HS-HS], [HS-LS], and [LS-LS] pairs, which change the magnetic situation within the respective molecule drastically. [71]

In continuation, such tetranuclear ST-Fe<sup>II</sup><sub>4</sub> units were used as functional bricks in the construction of more extended 1D and 2D nano-architectures. Thereby, the transition temperature of a 1D- columnar structure experienced a shift of ca. 200 K towards higher temperatures (compared to the unconnected ST-Fe<sup>II</sup><sub>4</sub> brick), while the 2D structure has been frozen in its LS state as consequence of the increased molecular rigidity in the extended 2D architectures (Figure 16). [72]



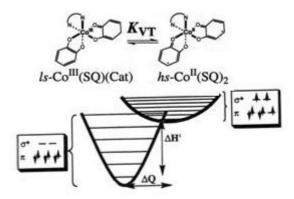
**Fig. 16:** Two-step hierarchical self-assembly of ST-module and of 1D- and 2D-architectures involving different types of metal ions (self-assembly I: Fe<sup>II</sup>; self-assembly II: La<sup>III</sup> or Ag<sup>I</sup>). [72]

Going towards infinite switching structures, the 1D coordination polymer of the composition [Fe<sup>II</sup>(triazole)<sub>3</sub>]<sub>n</sub> exhibits ST with a well developed hysteresis loop centred around room temperature. [73] Consequently, this type of compounds was used in first prototype applications, e. g. as thermo-reversible magneto-optical gels and as contrast agents in magnetic resonance imaging techniques. [74] [75]

More recent developments were directed towards the incorporation of spin transition centres into extended 3D-metal-organic frameworks. [76] The introduction of switchable magnetic ST bricks into metal-organic frameworks represents a fascinating approach towards new materials with multiple functionalities. Very recently, Kepert and co-workers reported on the successful formation of a cubic nanoporous 3D-networks out of ST-[Fe $^{II}_{2}$ (azpy)<sub>4</sub>(NCS)<sub>4</sub>] units (azpy = trans-4,4 $^{-}$ -azopyridine), which can host reversibly guest molecules as ethanol or methanol. While in the absence of absence of the guests, the host-network exhibits no ST, the inclusion of alcohol molecules induces ST. [77] A second example of 3D molecular solids exhibiting reversible hydration/dehydration-induced ST properties in the crystal was reported by Real et. al.. Here, controlled removing and adding of water (by warming or decrease of pressure) leads to a reversible shift of the critical temperatures and hysteresis width. Obviously, the compounds of the type [Fe(pmd)(OH)<sub>2</sub>{M(CN)<sub>2</sub>}<sub>2</sub>]. H<sub>2</sub>O (pmd = pyrimidine; M = Ag<sup>I</sup> or Au<sup>I</sup>) undergo a partially, but completely reversible ligand exchange at the ST-Fe<sup>II</sup> centres, whereby in detail the water molecules replace easily monodentate pyrimidine groups. [78]

## 5 Valence Tautomeric (VT) Compounds

The intramolecular phenomenon called Valence Tautomerism (VT) involves reversible electron transfer between the d-orbitals of a metal ion and a redoxactive ligand and might be accompanied simultaneously by an intra-orbital reorganisation of electron density (spin transition). VT has been reported for complexes containing a variety of metal ions (Mn, [79] Rh, and Ir [80]). However, most of scientific work has been recorded on the molecular system incorporating Co-ions ligated to dioxolene ligands. [81] So far, the main molecular Co-VT systems involve quinone-type ligands in their dianion form (catecholate, Cat) and in their radical anion form (semiquinone, SQ), respectively. The quinone ligand orbitals are generally in closed neighbourship to the metal orbitals and are so enabled to donate electrons to or to accept electrons from the metal. Additionally, most of the VT complexes undergo simultaneous changes in d-orbital occupation as a function of temperature. The LS-Co<sup>III</sup> form exhibits at the metal side an electronic configuration  $[\pi]^6(\sigma^*)^0$ , and the HS-Co<sup>II</sup>  $[\pi]^5(\sigma^*)^2$ , respectively (Figure 17). The occupation of the  $\sigma^*$  orbitals occurs only in the HS-Co<sup>II</sup> form and results in elongated Co-ligand bond lengths. The transition to longer bond lengths is accompanied by a large entropy increase due to the higher density of vibrational levels in the HS-Co<sup>II</sup> form.



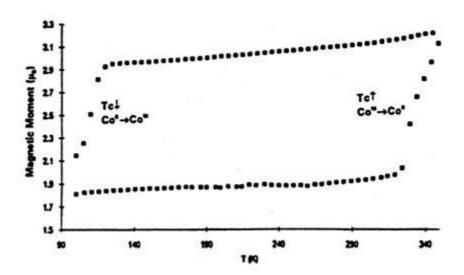
**Fig. 17:** The schematic representation and electronic configuration of molecular valence tautomerism (VT) (from [82]).

Following this description, the complex (N-N)LS-Co<sup>III</sup>(SQ)(Cat) is consequently transformed into the (N-N)HS-Co<sup>III</sup>(SQ)<sub>2</sub> complex by increasing the temperature. This corresponds basically to a spin transition and is defined by the low-spin ( $^6$  Co<sup>III</sup>) to high-spin ( $^7$  Co<sup>II</sup>) multiplicity change at the metal ion. Equally, the electron transfer involves either Co<sup>III</sup>-reduction or reduction of a SQ to Cat (by Co<sup>II</sup>).

Most of the VT-complexes reported to date are composed of, beside of the two dioxolene ligands and the cobalt ion, an additional aromatic N-N ligand (Figure 17). The most common dioxolene ligands are 3,5-di-*t*-butylsemiquinone (3,5-DBSQ), 3,6-di-*t*-butylsemiquinone (3,6-DBSQ), and their corresponding catecholate forms (3,5-DBCat and 3,6-DBCat), the N-N ligands bypyridine and phenantroline.

The magnetic properties of the two tautomeric forms are quite different and the transition process between them is sometimes companied of large hysteresis phenomena, as shown in Figure 18. Typical values of  $\chi T$ , the paramagnetic susceptibility product-temperature product,

range from 2.4 to 3.8 emu K mol<sup>-1</sup> (for the (N-N)HS-Co<sup>III</sup>(SQ)<sub>2</sub> form) and 0.4 emu K mol<sup>-1</sup> for the (N-N)LS-Co<sup>III</sup>(SQ)(Cat) form. In addition to the magnetic differences, the optical properties of the VT-tautomers differ substantially. Thus, the (N-N)LS-Co<sup>III</sup>(SQ)(Cat) tautomer is characterized by a band near 16500 cm<sup>-1</sup> (ca. 600nm) and a ligand-based intervalence transition near 4000 cm<sup>-1</sup> (2500 nm). On the other hand, the corresponding (N-N)HS-Co<sup>III</sup>(SQ)<sub>2</sub> tautomer shows a metal-to ligand charge transfer (MLCT) band near 13000 cm<sup>-1</sup> (770 nm), but no transition in the far-IR region. [83] The interconversion of the two tautomeric forms can be followed quite easily using variable-temperature electronic absorption spectroscopy. The transformation is marked by several isosbestic points, consistent with two species only, i.e. no further intermediates are involved. In similarity to the ST compounds, it was shown that (N-N)HS-Co<sup>III</sup>(SQ)<sub>2</sub> can be also interconverted reversibly into the (N-N)LS-Co<sup>III</sup>(SQ)(Cat) tautomer by application of pressure and light. [84] [85]



**Fig. 18:** Magnetisation *vs.* temperature diagram representing the very extended hysteresis loop and magnetic bistability for Co(3,6-DBSQ)<sub>2</sub>(PyO). [86]

### 6 Conclusion and Outlook

The now 10 years ongoing stormy development of the field of Molecular Magnetism has been rendered some important results as, among other examples, the emergence of purely organic magnetism and the realisation of room temperature magnets. However, in order to exploit the potential of molecular magnetism to a larger extent, the peculiarities of molecular magnets (in comparison to traditionally used magnetic materials) have to be investigated and exploited in further detail. Thereby, several challenges have to be solved indispensably in near future: On the field of infinite molecular networks, increased coercitivities have to be achieved to provoke first examples of substitution of conventional magnetic bulk materials in devices. Single Molecule Magnets (SMMs) and Single Chain Magnets (SCMs) will be only considered as alternatives for future technologies, if increased anisotropy barrier heights and especially much higher blocking temperatures can be realized. Furthermore, the investigation of real

single molecule magnetic behaviour (that means of the magnetic properties of really isolated single molecules) is still in its infancy and the problem of the implementation of such systems into device architectures has not even put onto the road map yet. Concerning the quantum behaviour, the question of the emergence of quantum coherence has not yet answered to the necessary evidence. The question for the single molecule magnetic behaviour of Spin Transition (ST) and Valence Tautomeric (VT) compounds has not even risen yet.

Concludingly, the, by its emerging history already very interdisciplinary interwoven, field of Molecular Magnetism will have to develop more intensely into cutting-edge collaborations between Physics and Chemistry in order to master the enormous challenge of magnetism of molecules within the nano-regime.

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